## The Melting Point and the Density of Neptunium Metal. A Micro Melting Point Apparatus for Metals<sup>1</sup>

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Neptunium metal was prepared on the milligram scale by reduction of NpF<sub>3</sub> with barium. The melting point was determined as  $640 \pm 1^{\circ}$ ; the density was found to be  $19.5 \pm 0.5$  g. cm.<sup>-3</sup>. An apparatus designed for the measurement of the melting points of 10 to 1000 microgram samples of metals is described.

In connection with the production of some high purity neptunium metal for calorimetry and other physico-chemical experiments it was desirable to determine the melting point and the density of the specimens used. Neptunium isotope of mass 237 was employed.

## Experimental

The neptunium metal was produced by the reduction of neptunium trifluoride. Inasmuch as several trifluoride preparations made from hydroxide precipitates proved to be only partially reducible to the metal, neptunium oxalate was selected as the starting material for metal production. The light green oxalate was precipitated upon addition of 1 M ammonium oxalate to a 1 M hydrochloric acid solution of the neptunium(IV) stock. After centrifugation and withdrawal of the supernatant liquid, the precipitate was washed with dilute oxalic acid and with water. The oxalate was slurried into a small platinum dish, placed in the platinum hydrofluorination apparatus,4 dried by passing anhydrous oxygen over it, and converted to dioxide by heating to 650°. After cooling, an equimolar mixture of oxygen and hydrogen fluoride was used to prepare the tetrafluoride; This was reduced with a 3:1 hydrogen-hydrogen fluoride mixture at a maximum temperature of 650°. The reduction to metal was achieved in a microfurnace and by a technique similar to that used by one of us<sup>6</sup> for plutonium metal pro-duction and described by Fried and Davidson<sup>6</sup> for their use in the preparation of neptunium metal. Barium vapor at 1300° was used as a reductant for the neptunium trifluoride. Both the substances were contained in a sintered beryllium oxide crucible system made from a nitric acid paste of beryllium oxide powder. Several milligrams of purified neptunium stock were available for these experiments and the reductions were made on a scale varying ments and the reductions were made on a scale varying from the order one-tenth to two milligrams. Spectro-chemical analysis typically detected the presence of the following cations in the metal: Be, 0.013; Ca, 0.02; Fe, 0.08; U, 0.2; mole per cent. Elements not detected but for which the limits of detection were fairly high were: Ba, 0.08; K, 1.8; Li, 0.8; and Ta, 0.2 mole per cent. The har, 0.08; K. 1.8; D. 0.8; and Ta, 0.2 mole per cent. The nearly spherical globules of metal produced were cleaned in a dry box under anhydrous xylene and sealed in evacuated glass tubes until used. The metal is silvery in color and has a malleability similar to that of uranium prepared on this scale.

Determination of the Np Melting Point.—In melting minute, irregularly shaped fragments of crystalline salts the deformation of the crystal under the surface tension of the liquid frequently serves as an indication of fusion. Alternatively, the disappearance upon fusion of the interference patterns of anisotropic substances observed with a polarizing microscope often provides a convenient indication of the change of phase of the material. Neither of these methods can be used on metallic globules. Furthermore, the formation of a very thin oxide film upon readily oxidized metals may so retain the metal that a bent disk may not collapse at the fusion temperature under the combined effect of gravity and the surface tension. The evidence of fusion may be postponed for tens of degrees above the fusion temperature for some actinide elements even in high vacuum.



Fig. 1.—Micro melting point apparatus.

The construction and operation of our micro apparatus for the determination of the melting point of microgram quantities of even readily oxidized metals may be seen from Fig. 1. The apparatus is contained in a Pyrex glass bulb (A) connected to a high vacuum line through (N). The (A) connected to a high vacuum line through (N). The sample and furnace mounted on 2 mm. diameter wolfram rods (C) spaced with nonex glass (B) are removable through the standard taper joint (P). The minute globule of metal (H) is held between the plane, parallel faces of the jaws of the 1 mm. diameter wolfram wire spring (E) by a force of the order of a gram weight. The spring is mounted with a tantal we applied surface of the plane, and surface of the spring is mounted with a tantal weight. the order of a gram weight. The spring is mounted with a tantalum clamp (D) and surrounded by a cylindrical tantalum radiation shield (M) 0.05 mm, thick and 2.2 cm, in diameter. A double Fibreglas insulated iron-constantan thermocouple (O) is spot welded to this spring near the sample at (G) (see insert I) and serves to measure the sample temperature. Observation of the specimen is made along the code (K) with a low rower binocular microscope through the axis (K) with a low power binocular microscope through the polished Pyrex optical port (J). The temperature of the radiation shield and its contents are controlled by means of the copper tubing coil (L) connected to a 300-watt portable oscillator to be described in a subsequent paper. In operation, the furnace and bulb are thoroughly outgassed at a temperature several hundred degrees higher than anticipated for the melting point and the vacuum is subsequently broken with purified argon. Then a small globule of metal is placed between the jaws of the wolfram spring with a jeweler's forceps with ground tips. To aid in placing and to prevent the loss of the sample in transfer to the apparatus the globule was occasionally coated with a small quantity of a volatile grease. Upon reassembling the apparatus and achieving a high vacuum the power output of the oscillator is gradually increased to heat the shield at the desired rate, but not beyond the ability of the pumping system to maintain a pressure about  $10^{-6}$  mm. despite the degassing of the hot metal parts. The thermocouple e.m.f. was bucked with a Rubicon Type B potentiometer and was recorded automatically on a Leeds and Northrup Speedomax re-corder with a precision of about 0.5°. The behavior of the sample was observed microscopically through an optical

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<sup>(4)</sup> E. F. Westrum, Jr., and L. Eyring, THIS JOURNAL, 78, 3396 (1951).

<sup>(5)</sup> E. F. Westrum, Jr., Manhattan Project Reports (CK-1586) May, 1944; E. F. Westrum, Jr., Manhattan Project Reports (CN-2495) Jan., 1945.

<sup>(6)</sup> S. Fried and N. Davidson, THIS JOURNAL, 70, 3539 (1948).

window sealed into the bulb. The observation of fusion (convergence of the jaws) was either relayed or recorded with a magnetically activated pen on the Speedomax recorder.

In addition to using calibrated thermocouple wire, the apparatus was checked against small pieces of National Bureau of Standards freezing point samples of magnesium and zinc with stated freezing points of 651 and 419.5°. We observed melting point values of 650 and 419.5°, respectively.

Two independently prepared samples of neptunium metal of about 200 micrograms each yielded melting point values of 639 to 641 and 639°. The metal showed no tendency to adhere to the wolfram jaws but formed a thin circular foil between them with the bulk of the metal extruding out and forming a bright sphere adjacent to the jaw opening. This behavior made reloading of the same sample convenient and facilitated repeated determinations. A sharp convergence of the jaws was noted for all the samples of Np, Zn and Mg.

In adapting the original apparatus for use in work involving thorium metal and protactinium compounds, Dr. Walter Blaedel and the authors modified the apparatus by use of a platinum versus platinum-10% rhodium thermocouple and a 2.5 kw. oscillator to attain temperatures in excess of 1300°. For still higher temperatures a helical wolfram wire heater was substituted for the radiation shield and a Leeds and Northrup optical pyrometer used to determine the sample temperature after appropriate correction for the adsorption by the glass port and the emissivity of the wolfram surface. Good agreement with the best reported values of the thorium melting point were obtained.

Surface. Used agreement which the less reported values of the thorium melting point were obtained. **The Density of Neptunium Metal.**—Fried and Davidson<sup>3</sup> reported densities of 17.9 and 17.6 g. cc. <sup>-1</sup> on a forty-microgram piece of metal in an apparatus described by Kirk. We have determined the density of a number of specimens of Np metal by a similar method (devised by S. Katz) in which a precision bore capillary tube with polished plane sides parallel to the axis and a scribed fiducial mark is employed. The displacement of a dibutyl phthalate meniscus upon the immersion of the metal sample in the capillary and the centrifugation of the entire tube was used to determine the volume of the metal sample. Two capillaries of 0.5 and 0.75 mm. diameter were calibrated both by optical measurement of the diameter and by experiments using pellets of platinuun of a density of 21.4 g. cm.<sup>-3</sup>. The two methods gave agreement within about 2%.

## Results

The results of the density measurements are summarized in the Table I.

Each determination represents metal from a different metal preparation and the figures reported are in general the mean value of repeated experi-

TABLE I Density of Neptunium Metal

Np sample	Wt. of Np (mg.)	Density (g./cc.)
1	0.425	17.4,16.9
2	1.553	<b>19.7,2</b> 0.1,19.4
3	1.774	19.4, 19.3
4, 5, 6 (3 pieces)	0.685	19.2

ments. The low density of sample 1 (which is in good agreement with the values of Fried and Davidson) may be due to the occurrence of a higher temperature metallic phase stabilized by the presence of an impurity. We believe that the higher densities  $(19.5 \pm 0.5)$  found in the subsequent experiments may represent the normal density of the phase stable at room temperature.

Unsuccessful attempts were made to change sample one to a higher density form. Sample four was annealed for two hours at 90°, whereas samples five and six had no treatment beyond metal production. X-Ray examination by Dr. David Templeton of all three samples (four, five and six) yielded identical X-ray patterns having a facecentered cubic structure with a cell constant of  $4.89^{\circ}$  K. and a calculated density of 14.7 g. cm.<sup>-3</sup>. The pattern is attributed to a thin oxide film rather than the bulk metal itself. A microdilatometric experiment on larger quantities of neptunium now available would be of interest in establishing the presence of the suspected transitions above room temperature. An additional indication of polymorphism is the fact that the originally bright, smooth, metallic surface of the globule extruded in the melting experiment was observed to wrinkle on cooling to approximately 100° suggesting a transition involving a considerable volume change. The observed wrinkling is not, of course, a reversible effect.

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